

What high pressure studies have taught us

What High Pressure Studies Have Taught Us About High-Temperature Superconductivity

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Abstract

Superconductivity is an important area of modern research which has benefited enormously from experiments under high pressure conditions. The focus of this paper will be on three classes of high-temperature superconductors: (1) the new binary compound MgB_2 , (2) the alkali-doped fullerenes, and (3) the cuprate oxides. We will discuss results from experiment and theory which illustrate the kinds of vital information the high-pressure variable can give to help better understand these fascinating materials.

Key words: high pressure, fullerene superconductors, oxide superconductors, MgB_2 superconductor, superconductivity

1. INTRODUCTION

Superconductivity was discovered in 1911 by G. Holst and Kamerlingh Onnes in Leiden when elemental Hg was cooled to temperatures below 4.15 K [1]. Fourteen years later the same group carried out the first high-pressure experiments on a superconductor [2]. As reported by D. H. Bowen in his review paper [3]: “The first experiments in which stresses were deliberately applied to superconductors were by Sizoo and Onnes in 1925 who found that the transition temperatures of tin and indium were increased by uniaxial tension and decreased by applying pressure to the helium bath in which they were immersed”. In the 76 years since these first experiments, high-pressure experiments have had a considerable impact on

the field of superconductivity. This is not surprising, since the application of high pressures has:

1. led to the discovery of many new superconductors, including 22 elemental solids
(B, O, Si, P, S, Ca, Sc, Fe, Ge, As, Se, Br, Sr, Y, Sb, Te, I, Cs, Ba, Bi, Ce, Lu) bringing the total number of elemental-solid superconductors to 51.
2. aided in the synthesis of novel high quality superconducting materials. This the subject of a paper by C.W. Chu at this conference and will not be discussed here.
3. guided efforts to enhance the transition temperature T_c through chemical means. Even without a detailed understanding of *why* T_c changes with pressure, a large magnitude of the pressure derivative dT_c/dP is a good indication that higher values of T_c are possible at ambient pressure through chemical means. The observation of a large enhancement in T_c under pressure in the high-temperature oxide superconductor La-Ba-Cu-O prompted the substitution of the smaller ion Y^{3+} for La^{3+} to generate lattice pressure, leading to the important discovery [4] of superconductivity in $YBa_2Cu_3O_{7-\delta}$ (Y-123) at 92 K. Below we will see that efforts are being made to expand the lattice of hole-doped C_{60} in hopes of raising its $T_c \simeq 52$ K to even higher values [5].
4. yielded the dependence of T_c on sample volume and lattice parameters which helped identify the pairing mechanism and test theoretical models.

This paper will focus its attention on this fourth “benefit” of high pressure research. The types of superconductor that we consider are the binary compound MgB_2 , electron- and hole-doped fullerenes, and the cuprate oxides. These materials owe their extraordinary superconducting properties to their reduced (2D) dimensionality. The electron pairing leading to superconductivity takes place primarily within the B_2 -layers in MgB_2 , within the CuO_2 -planes in the oxides, and on the surface (2D!) of the large C_{60} molecule in the doped fullerenes.

The results of high pressure experiments on these important materials can be best understood if we first consider analogous experiments on conventional simple metal and transition metal superconductors.

2. SUPERCONDUCTIVITY IN CONVENTIONAL SUPERCONDUCTORS

The fact that high pressure creates 22 new elemental superconductors does *not* imply that superconductivity is normally enhanced under pressure; in fact, just the opposite is true. The majority of the above “high-pressure superconductors” entered this state following a pressure-induced insulator→metal transition.

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For most known superconductors, T_c decreases under pressure, sometimes quite rapidly; a positive value of dT_c/dP is rather infrequent. The reason for this can be most easily understood by considering the simple metal superconductors, like Sn, In, Pb, and Al, where the conduction electrons possess s, p character. In all simple metal superconductors, dT_c/dP is negative [6]: for example, Sn (-0.482 K/GPa), In (-0.381 K/GPa), and Pb (-0.365 K/GPa). This ubiquitous decrease in T_c is not an electronic effect, but arises predominantly from a stiffening of the lattice with increasing pressure, the same reason the equation-of-state $V(P)$ has an upward (positive) curvature!

We can make these arguments more specific by considering the BCS expression for the transition temperature

$$T_c \simeq \langle \omega \rangle \exp \left\{ \frac{-1}{N(E_f) \mathcal{V}} \right\}, \quad (1)$$

where $\langle \omega \rangle$ is an average lattice-vibration frequency, $N(E_f)$ is the electronic density of states at the Fermi energy, and \mathcal{V} is the attractive pairing interaction. Since the s, p electrons in simple metals are nearly free, one expects approximately $N(E_f) \propto V^{+2/3}$ so that $N(E_f)$ decreases even more slowly than the sample volume V with increasing pressure. However, the principal reason for the observed decrease in T_c with pressure is that the pairing interaction \mathcal{V} itself decreases by a sizeable amount due to lattice stiffening, which makes it increasingly difficult for the anions in the crystal lattice to couple with the electrons.

To put this discussion on a more quantitative basis, we need to consider the McMillan equation [7]

$$T_c \simeq \frac{\langle \omega \rangle}{1.20} \exp \left\{ \frac{-1.04(1 + \lambda)}{\lambda - \mu^*(1 + 0.62\lambda)} \right\}, \quad (2)$$

which connects the value of T_c with fundamental parameters such as the electron-phonon coupling parameter λ , an average phonon frequency $\langle \omega \rangle$, and the Coulomb repulsion which we set equal to $\mu^* = 0.1$. The coupling parameter itself is defined by $\lambda = N(E_f) \langle I^2 \rangle / [M \langle \omega^2 \rangle]$, where $\langle I^2 \rangle$ is the average squared electronic matrix element, M the molecular mass, and $\langle \omega^2 \rangle$ the average squared phonon frequency. Taking the logarithmic volume derivative of T_c in Eq. (2), we obtain the simple relation

$$\frac{d \ln T_c}{d \ln V} = -B \frac{d \ln T_c}{dP} = -\gamma + \Delta \left\{ \frac{d \ln \eta}{d \ln V} + 2\gamma \right\}, \quad (3)$$

where B is the bulk modulus, $\gamma \equiv -d \ln \langle \omega \rangle / d \ln V$ the Grüneisen parameter, $\eta \equiv N(E_f) \langle I^2 \rangle$ [8], and $\Delta \equiv 1.04\lambda[1 + 0.38\mu^*][\lambda - \mu^*(1 + 0.62\lambda)]^{-2}$. Eq. (3) has a simple interpretation. The first term on the right, which comes from the prefactor to the exponent in the above McMillan expression for T_c , is usually small relative to the second term. The sign of the pressure derivative dT_c/dP , therefore, is determined by the relative magnitude of the two terms in the curly

brackets.

The first “electronic” term in the curly brackets involves the derivative of the Hopfield parameter $\eta \equiv N(E_f) \langle I^2 \rangle$, an “atomic” property which can be calculated directly in band-structure theory. In his landmark paper [7], McMillan demonstrated that whereas individually $N(E_f)$ and $\langle I^2 \rangle$ may fluctuate appreciably, their product $\eta \equiv N(E_f) \langle I^2 \rangle$ changes only gradually, i.e. η is a well behaved “atomic” property. One would thus anticipate that η changes in a relatively well defined manner under pressure, reflecting the character of the electrons near the Fermi energy. An examination of the body of high-pressure data on simple s, p -metal superconductors, in fact, reveals that η normally increases under pressure at a rate close to $d \ln \eta / d \ln V \approx -1$ [9]. For transition-metal (d -electron) superconductors, on the other hand, Hopfield has pointed out [8] that the larger value $d \ln \eta / d \ln V \approx -3$ to -4 is more appropriate.

Let us now apply Eq. (3) to an analysis of dT_c/dP for simple metal superconductors. The second “lattice” term in the curly brackets in Eq. (3) is positive since the lattice term is positive ($2\gamma \approx +3$ to $+5$) and dominates over the negative electronic term $d \ln \eta / d \ln V \approx -1$. Since Δ is always positive and $-\gamma$ is relatively small, the sign of dT_c/dP is negative, opposite to that in the curly brackets. This accounts for the universal decrease of T_c with pressure due to lattice stiffening in simple metals. In Sn, for example, T_c decreases under pressure at the rate $dT_c/dP \simeq -0.482$ K/GPa which leads to $d \ln T_c / d \ln V \simeq +7.2$ [6]. Inserting for Sn $T_{c0} \simeq 3.73$ K, $\langle \omega \rangle \simeq 110$ K [10], and $\mu^* = 0.1$ into the above McMillan equation, we obtain $\lambda \simeq 0.69$ from which follows that $\Delta \simeq 2.47$. Inserting the above values into Eq. (3) and setting $d \ln \eta / d \ln V \approx -1$, we can solve Eq. (3) for the Grüneisen parameter to obtain $\gamma \simeq +2.46$, in reasonable agreement with the experimental value $\gamma \approx +2.1$ [6]. Similar results are obtained for other conventional simple metal BCS superconductors.

In transition metal superconductors the electrons taking part in the superconductivity have predominantly d character which often leads to higher values of the density of states $N(E_f)$ and transition temperature T_c . In many transition metals T_c decreases with pressure, but in some T_c increases. Indeed, under pressure $N(E_f)$ can either decrease or increase; should E_f lie on the low energy side of a peak in $N(E)$, $s \rightarrow d$ electron transfer under pressure would lead to an increase in $N(E_f)$, and vice versa should E_f lie on the high energy side of a peak. The moderating influence of the change in $\langle I^2 \rangle$ under pressure leads to the universal increase of their product $\eta \equiv N(E_f) \langle I^2 \rangle$ according to $d \ln \eta / d \ln V \approx -3$ to -4 , as pointed out by Hopfield [8]. If this relatively large electronic term becomes larger than the lattice term 2γ in Eq. (3), T_c would be expected to *increase* with pressure; this is, in fact, observed in the transition metals V [11] and La [12], among others. Unlike for s, p metals, the pressure dependence $T_c(P)$ for transition metals follows no universal behavior, reflecting the additional complexity, and potency, of the electronic properties in a d electron system.

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3. SUPERCONDUCTIVITY IN MgB_2

The discovery of superconductivity at the high temperature $T_c \approx 40$ K in the simple s, p -metal compound MgB_2 was quite unexpected [13]. The absence [14] of the problematic weak-link behavior of the high- T_c oxides and the relative ease of synthesis in various forms [15] has raised hopes that MgB_2 may be suitable for numerous technological applications. Efforts to enhance the value of T_c in this class of superconductor would be aided by the identification of the superconducting mechanism and by establishing systematics in the superconducting and normal-state properties. MgB_2 is a quasi-2D material with strong covalent bonding within the boron layers. It is thus not surprising that the compression is anisotropic [16–18, 27], the most accurate structural measurements [27] revealing that under hydrostatic pressure the initial compression along the c axis is 64% greater than along the a axis; the bulk modulus is $B = 147.2(7)$. The anisotropy in the superconducting properties is also appreciable, the upper critical field ratio $H_{c_2}^{ab}/H_{c_2}^c$ reportedly being 2 – 3 [19, 20], less than that observed in the high- T_c oxides [21]. A full characterization of the remaining anisotropic properties awaits the synthesis of sufficiently large single crystals.

Soon after the discovery of superconductivity in MgB_2 , three groups [22–24] reported that T_c decreased under high pressure, but the rate of decrease varied considerably from -1.6 K/GPa [22] to -1.9 K/GPa [23] in piston-cylinder studies with Fluorinert pressure medium, to \sim -0.6 K/GPa in quasi-hydrostatic studies [24] to 25 GPa with solid steatite pressure medium. Choi *et al.* [25] have recently carried out resistivity studies to 1.5 GPa pressure in daphne-kerosene pressure medium, obtaining $dT_c/dP \simeq -1.36$ K/GPa. The differing pressure dependences may be due to differences in the samples and/or to shear stress effects in the frozen or solid pressure media.

We recently carried out He-gas hydrostatic pressure experiments [26] on the same high quality isotopically pure (^{11}B) sample used in the above structural studies [27]. For $P \lesssim 0.5$ GPa helium is fluid at $T_c \approx 39$ K. At higher pressures the shear stresses are held to a minimum by the carefully controlled manner [28] in which solid helium is allowed to freeze around the sample. The dependence of T_c on hydrostatic pressure is seen in Fig. 1(a) to be highly linear, $dT_c/dP \simeq -1.11(2)$ K/GPa (yielding $d \ln T_c / d \ln V = B d \ln T_c / dP \simeq +4.16$), and does not depend on the pressure/temperature history of the sample. Such history effects are rare in superconductors without pressure-induced phase transitions, but do occur in certain high- T_c oxides containing defects with appreciable mobility at RT [29]. In addition, we observed that dT_c/dP remained unchanged if neon was substituted for He as pressure medium, confirming the absence of intercalation effects in MgB_2 . Lorenz *et al.* [30] have very recently carried out He-gas studies to 0.8 GPa on two MgB_2 samples with differing T_{c0} values 39.2 K and 37.5 K obtaining $dT_c/dP \simeq -1.07$ K/GPa and -1.45 K/GPa, respectively. These authors conclude that differences in the samples themselves, and not shear

stress effects, are responsible for the differing dT_c/dP values.

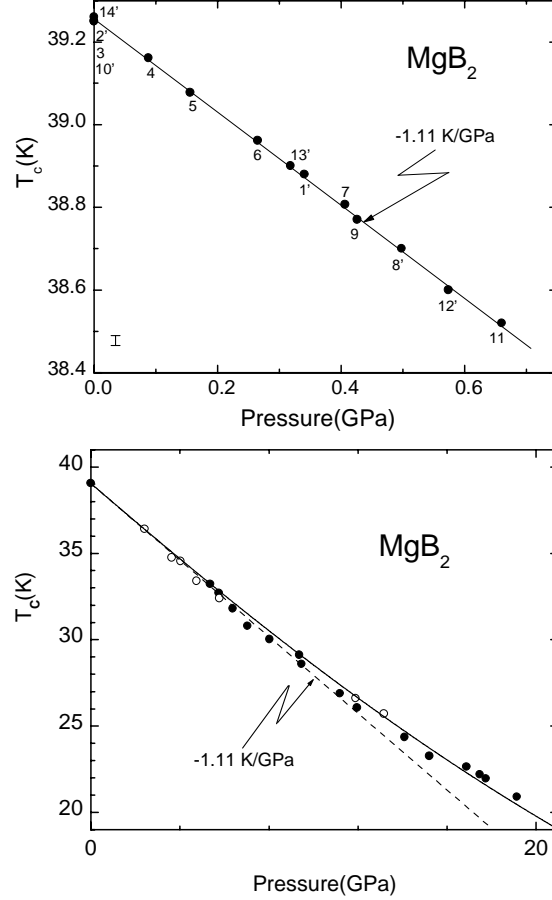


Fig. 1. (a)(top) Superconducting transition temperature onset versus applied pressure from Ref. [26]. Numbers give order of measurement. A typical error bar for T_c (± 0.01 K) is given in lower left corner; the error in pressure is less than the symbol size. Pressure was either changed at RT (unprimed numbers) or at low temperatures ~ 60 K (primed numbers). (b)(bottom) Superconducting transition temperature midpoint versus pressure to 20 GPa from diamond-anvil-cell measurements in Ref. [31]. Data taken first for monotonically increasing pressure (\bullet), then for monotonically decreasing pressure (\circ). The straight dashed line has slope -1.11 K/GPa.

In Fig. 1(b) we show the dependence of T_c on pressure to 20 GPa for MgB_2 using a diamond-anvil-cell (DAC) with dense helium pressure medium [31], thus extending the pressure range of the above He-gas studies nearly thirtyfold. T_c is seen to decrease approximately linearly with pressure to 10 GPa, consistent with

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the rate -1.11 K/GPa (dashed line). Very recently Tissen *et al.* [32] have carried out ac susceptibility measurements in a DAC to 28 GPa on a MgB₂ sample with $T_{c0} \simeq 37.3$ K at ambient pressure. They find the high initial slope $dT_c/dP \simeq -2$ K/GPa, T_c decreasing to 11 K at 20 GPa and 6 K at 28 GPa. They also report that the pressure dependence $T_c(P)$ shows a bump near 9 GPa which they speculate may arise from an electronic Lifshitz transition. Further experiments are necessary to determine whether this bump is intrinsic to the sample or the result of shear stress effects in the frozen methanol-ethanol pressure medium.

We now apply the same analysis carried out above for simple metal superconductors to MgB₂ to see whether $T_c(P)$ from our measurements is consistent or not with BCS electron-phonon coupling theory. We first analyze the He-gas data which gives the initial dependence. Using the average phonon energy from inelastic neutron studies [33] $\langle\omega\rangle = 670$ K, $T_{c0} \simeq 39.25$ K, and $\mu^* = 0.1$, we obtain from the above relations $\lambda \simeq 0.90$ and $\Delta \simeq 1.75$. Our estimate of $\lambda \simeq 0.90$ agrees well with those of other authors [34, 35]. Since the pairing electrons in MgB₂ are believed to be *s, p* in character [35–38], we set $d \ln \eta / d \ln V \approx -1$, a value close to $d \ln \eta / d \ln V = B d \ln \eta / d P \approx -0.81$, where $B = 147.2$ GPa from Ref. [27] and $d \ln \eta / d P \approx +0.55$ %/GPa from first-principles electronic structure calculations by Medvedera *et al.* [39]. Inserting the above values of $d \ln T_c / d \ln V = +4.16$, $\Delta = 1.75$, and $d \ln \eta / d \ln V = -1$ into Eq. (3), we find $\gamma = 2.36$, in reasonable agreement with the value $\gamma \approx 2.9$ from Raman spectroscopy studies [18] or $\gamma \approx 2.3$ from *ab initio* electronic structure calculations on MgB₂ [40]. The He-gas $T_c(P)$ data are thus clearly consistent with electron-phonon pairing in MgB₂.

A more stringent test of this conclusion is provided by the DAC $T_c(P)$ data in Fig. 1(b) which cover a relatively large $\sim 10\%$ change in volume. We would like to see whether we can reproduce the DAC data using the McMillan equation and suitably extrapolating the parameters used in the above analysis of the He-gas data. As pointed out by Chen *et al.* [41], an appropriate method of extrapolation is to integrate the volume derivatives of the above parameters $\gamma \equiv -d \ln \langle\omega\rangle / d \ln V = +2.36$ and $d \ln \lambda / d \ln V = d \ln \eta / d \ln V - d \ln \langle\omega^2\rangle / d \ln V = -1 - 2(-2.36) = +3.72$ to obtain $\langle\omega\rangle = (670 \text{ K})(V/V_0)^{-2.36}$ and $\lambda = 0.90(V/V_0)^{3.72}$. Inserting these two volume dependences in the McMillan equation, and assuming $\mu^* = 0.1$ is independent of pressure [41], we obtain the dependence of T_c on relative volume V/V_0 . This can be converted to the dependence of T_c on pressure P by using the Murnaghan equation-of-state $V(P)/V_0 = [1 + B'P/B]^{-1/B'}$ where we use the value $B = 147.2$ GPa from Ref. [27] and the canonical value $B' \equiv dB/dP = 4$ supported by a recent calculation [42]. As seen in Fig. 1(b), the agreement of this calculated $T_c(P)$ dependence (solid line) with the experimental data is quite impressive. According to this estimate, a pressure of $P \approx 50$ GPa would be required to drive T_c to below 4 K. A similar calculation was very recently carried out by Chen *et al.* [41] over a much wider pressure range; this paper also contains a detailed discussion

of the pressures dependences of η , λ , and μ^* .

The good agreement between the experimental data to 20 GPa and the predictions of the McMillan formula using the volume dependences determined from the He-gas high-pressure data to 0.7 GPa provides substantial evidence that superconductivity in MgB₂ originates from standard BCS phonon-mediated electron pairing. This finding agrees with high precision isotope effect experiments [43,44], among others. The fact that the B isotope effect is fifteen times that for Mg [44] is clear evidence that the superconducting pairing originates within the graphite-like B₂-layers

4. SUPERCONDUCTIVITY IN THE ALKALI-DOPED FULLERENES

Another class of superconductors having very high values of T_c are the alkali-doped fullerides A₃C₆₀, where A = K, Rb, Cs, or some combination thereof [45]. As for the simple metal and MgB₂ superconductors, T_c for the alkali-doped fullerides is found to decrease under the application of hydrostatic pressure [46]. For example, for Rb₃C₆₀, where $T_{c0} \simeq 29.5$ K, we obtained $dT_c/dP \simeq -8.7$ K/GPa [47]. From our measurement of the bulk modulus in a neutron diffraction experiment [48], $B = 18.3$ GPa, we estimated $d \ln T_c / d \ln V = B d \ln T_c / dP \simeq +5.4$, a value intermediate between that for MgB₂ (+4.16) and that for Sn (+7.2). One is tempted to account for the decrease in T_c with pressure for Rb₃C₆₀ or the other A₃C₆₀ fullerides by invoking pressure-induced lattice stiffening. Such an attempt, however, fails. In contrast to MgB₂ and Sn, the electronic bandwidth is quite narrow, leading to the expectation of a sizeable decrease in $N(E_f)$ under pressure. To determine the reason for the large negative value of dT_c/dP , we measured in a single experiment the pressure dependence of both T_c and $N(E_f)$ for Rb₃C₆₀, determining in the process the functional dependence of T_c on $N(E_f)$ [47]. A detailed analysis revealed that weak-coupling theory can account for these pressure dependences as long as the characteristic energy of the intermediary boson is between 300 K and 800 K, typical energies for the high frequency on-ball phonons. The mechanism behind the large negative value of dT_c/dP in Rb₃C₆₀ is thus not lattice stiffening, as in the simple metals and MgB₂, but a sharp decrease in the electronic density of states $N(E_f)$ with pressure.

This result sheds light on the observed increase in T_c in A₃C₆₀ as the larger Rb⁺¹ ion is substituted for the smaller K⁺¹, thus expanding the lattice, as seen in Fig. 2. In fact, it was widely believed that the relationship between T_c and the lattice parameter a followed a universal behavior in both cation substitution and high-pressure experiments. Our combined equation-of-state and $T_c(P)$ studies on Rb₃C₆₀, however, revealed that two different dependences are found, as seen in Fig. 2. The reason for this effect is not known, but may have to do with different rotationally ordered states of the C₆₀ molecule in the two cases which

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influences the density of states and thus T_c .

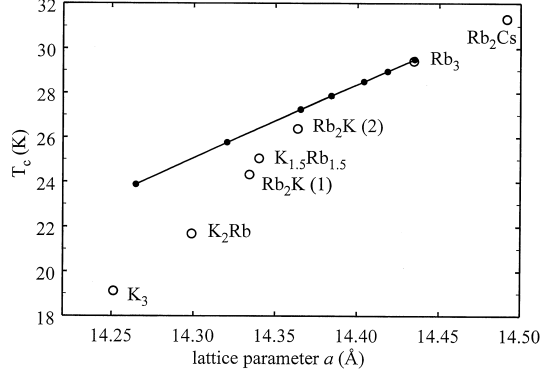


Fig. 2. T_c versus lattice parameter from Ref. [48] for both (●) high-pressure experiments on Rb_3C_{60} and (○) cation-substitution experiments. Only the cation(s) to C_{60} are listed in the figure. Solid line is guide to eye.

Using gate-induced doping in a field-effect transistor configuration, Schön *et al.* have demonstrated superconductivity in both electron- [49] and hole-doped [50] C_{60} single crystals with lattice parameter 14.16 Å. The maximum values of T_c are found to be 11 K and 52 K, respectively. The latter value is a record for a non-cuprate superconductor. The higher value of T_c for hole doping compared to electron doping is apparently correlated with the higher density of states in the valence band (HOMO) compared with the conduction band (LUMO).

From the above discussion, $T_c \simeq 52$ K for optimally hole-doped C_{60} would evidently be expected to increase further if its lattice parameter a could be expanded. We found above for Rb_3C_{60} that $d \ln T_c / d \ln V = d \ln T_c / 3 d \ln a \simeq +5.4$, which implies that $d \ln T_c / d \ln a \simeq +16.2$. This means that one would expect T_c to increase about 16 times faster than the lattice parameter a . If we expand the lattice parameter a of C_{60} by 2%, so that it equals that of Rb_3C_{60} , one would expect T_c to increase by approximately $16.2 \times 2\% \simeq 32\%$, implying an increase in T_c from 52 K to nearly 70 K. To test the above, it would be of interest to expand C_{60} 's lattice, perhaps through suitable intercalation of rare gas atoms.

5. SUPERCONDUCTIVITY IN THE CUPRATE OXIDES

It has been nearly 15 years since the discovery [51] of high- T_c superconductivity in the cuprate oxide La-Ba-Cu-O at $T_c \approx 35$ K. In spite of enormous efforts since then, including over 50,000 experimental and 20,000 theoretical papers, there is still no consensus on the underlying mechanism responsible for the superconducting pairing. This is one of the great unsolved problems in Condensed

Matter Physics. The maximum value of T_c has improved from initially ~ 35 K for La-Ba-Cu-O and related systems to 134 K for $\text{HgBa}_2\text{Ca}_2\text{Cu}_3\text{O}_{8+\delta}$ (Hg-1223) at ambient pressure in 1993 [52] and ~ 160 K for the same compound at 30 GPa pressure in 1994 [53]. There have even been unconfirmed reports of superconductivity at 250 K in $\text{Tl}_2\text{Ba}_2\text{Ca}_2\text{Cu}_3\text{O}_{10+\delta}$ (Tl-2223) above 4 GPa [54] and at 330 K in the Pb-Ag-C-O system at ambient pressure [55].

Unfortunately, Nature has been particularly inventive in her efforts to thwart understanding. The distorted modified perovskite structure of these materials undergoes subtle structural changes when temperature or pressure is varied which complicates the interpretation of experimental data. Defects of many different kinds have a profound influence on the superconducting properties, including the value of T_c . An example is the emergence of the low-temperature-tetragonal (LTT) phase in $\text{La}_{2-x}\text{Ba}_x\text{CuO}_4$ for $x \approx 0.125$ which completely suppresses T_c from its nominal value of ~ 37 K. Because of the strong inverse parabolic dependence of T_c on the carrier concentration n seen in Fig. 3 [56], small changes in n due to the influence of defects can have a surprising large effect on T_c .

In spite of the complexity of these materials, researchers [57] have been able to identify a number of important guidelines for enhancing the value of T_c in the superconducting oxides: (1) the carrier concentration n in the CuO_2 planes should be varied through suitable cation substitution until its optimal value is reached (see Fig. 3); (2) try to increase the number of CuO_2 planes which lie close together (in a packet) in the oxide structure while maintaining optimal doping - “respectable” one-plane systems, like Tl-2201, have T_c values in the range 90 - 100 K, two-plane systems in the range 100 - 120 K, and three-plane systems in the range 120 - 140 K; (3) try to position defects as far from the CuO_2 planes as possible; and (4) since T_c is diminished with increasing buckling angle in the CuO_2 planes, develop structures where the CuO_2 planes are as flat as possible.

We would like to pose the following question: are there perhaps other guidelines to maximize T_c which high pressure studies can give us? Can high pressure experiments give us NEW information not available from other studies? The most studied superconducting property under pressure is $T_c(P)$ which depends on the system studied, the doping level n , the type and mobility of defects, and, in some cases, on the pressure medium used [9, 58]. However, considering the totality of $T_c(P)$ data, one feature stands out: that more often than not T_c increases with pressure, as first recognized by Schirber [59]. The foregoing discussion should help the reader appreciate that an increase in T_c with pressure is something special! In the case of the optimally doped Hg compounds with one, two and three CuO_2 planes, the initial rate of increase is identical, $dT_c/dP \approx +1.75$ K/GPa [60]. It is this constant but perservering increase in T_c with pressure which allows T_c in Hg-1223 to increase from 134 K at ambient pressure to ~ 160 K at 30 GPa at which pressure $T_c(P)$ passes through a maximum [53].

In the cuprate oxides, the canonical change in T_c with pressure is that it first increases with pressure, passes through a maximum value at some critical

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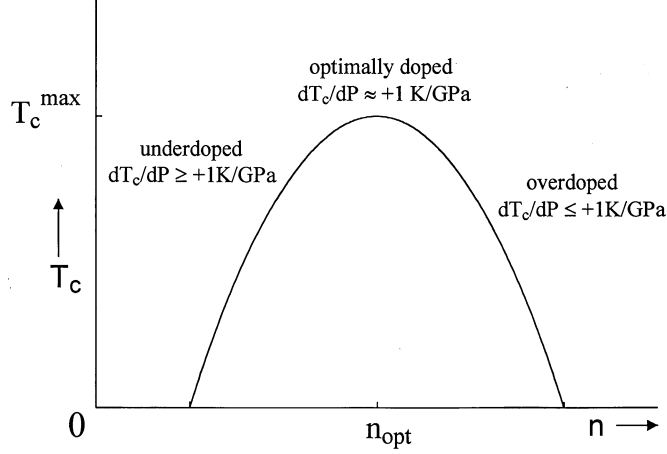


Fig. 3. Inverted parabolic dependence of T_c on hole-carrier content for the superconducting oxides adapted from Ref. [56]. Representative experimental values of dT_c/dP for underdoped, optimally doped and overdoped samples are given.

pressure, and then decreases [9, 58]. What causes this $T_c(P)$ dependence? An examination of Fig. 3 leads one to the idea that perhaps the hole-carrier concentration n increases with pressure so that for an underdoped sample, where $n < n_{opt}$, $T_c(P)$ simply tracks the canonical bell-shaped $T_c(n)$ curve. Indeed, Hall effect studies show that n normally increases with pressure at the rate $+10\%/GPa$ [9]. If this were the whole story, then for an optimally doped sample one would expect $dT_c/dP = (dT_c/dn)(dn/dP) = 0$, since $T_c(n)$ is at an extremum for $n = n_{opt}$ (see Fig. 3). As indicated in Fig. 3, however, this expectation from this simple charge-transfer model is not confirmed in experiment. For “normal” high- T_c oxides, $dT_c/dP \approx +1$ to $+2$ at optimal doping! This means that there are at least two effects determining the total pressure dependence of T_c

$$\frac{dT_c}{dP} = \left(\frac{dT_c}{dP}\right)^{intrinsic} + \left(\frac{dT_c}{dn}\right)\left(\frac{dn}{dP}\right), \quad (4)$$

an intrinsic dependence $(dT_c/dP)^{intrinsic} \approx +1$ to $+2$, and a second dependence arising from pressure-induced changes in the carrier concentration n in the CuO_2 planes. It is this intrinsic term which promises to tell us something new about high- T_c superconductors, something we perhaps could not have learned from ambient pressure experiments under varying sample stoichiometries and structure types. However, we have only evaluated $(dT_c/dP)^{intrinsic}$ at optimal doping. It is certainly possible that this intrinsic dependence varies with the carrier concentration. To extract the intrinsic term from $T_c(P)$ data with varying values of n , we must be able to estimate the second term on the right in Eq. (4), the charge

transfer term. This is no mean feat since this term depends on both the rate of change of n with pressure, as well as dT_c/dn .

There are, however, further problems which make the extraction of the intrinsic term even more difficult. In many superconducting oxides the application of pressure doesn't simply compress the lattice, but also prompts mobile oxygen defects to assume a greater degree of local order [9, 61]. This leads to relaxation effects which are both temperature and pressure dependent. The oxygen chain sublattice in $\text{YBa}_2\text{Cu}_3\text{O}_{7-y}$ (Y-123), for example, is partially occupied with oxygen anions with a considerable mobility at room temperature. This allows these oxygen defects to assume a myriad of different substructures or local ordered states which can influence the value of T_c . Under pressure the oxygen defects may migrate from one local ordered state to another, but at a progressively slower rate, the higher the pressure [62]. In fact, one can measure the pressure dependence of the relaxation time $\tau(P)$ and use this to estimate the most likely diffusion path of oxygen defects through the solid [63]. For studies of the $T_c(P)^{\text{intrinsic}}$, however, these relaxation effects are more than just a nuisance. One strategy to eliminate the relaxation effects completely is to carry out the entire high pressure experiment at low temperatures, allowing τ , which depends exponentially on temperature, to become extremely large. The hope would be that the elimination of relaxation effects would turn the complex dependency $T_c(n, P)$ into a simple one. Unfortunately, experiments to 20 GPa in a DAC on Y-123 carried out solely at temperatures low enough (< 90 K) to suppress oxygen ordering brought little simplification [62]. In the Y-123 system, at least, the extraction of the dependence $T_c(P)^{\text{intrinsic}}$ would seem out of reach.

Fortunately, these relaxation effects operate by changing the carrier concentration n . By considering pressure derivatives dT_c/dP only for optimally doped samples, we can eliminate not only any changes in T_c from the normal increase in n with pressure, but also those changes arising from oxygen ordering phenomena. Perhaps the most important fundamental result from all high pressure experiments on the optimally doped oxides is that $dT_c/dP = (dT_c/dP)^{\text{intrinsic}} \approx +1.5$ K/GPa which corresponds to the volume dependence $T_c \propto V^{-1.2}$ or $d \ln T_c / d \ln V \simeq -1.2$ [9], a far weaker volume dependence than obtained for the s, p metal systems Sn (+7.2) or MgB_2 (+4.16) where T_c increases with pressure. This comparison emphasizes how small the percent change in T_c under pressure really is for the superconducting oxides. The fact that numerous different systems yield approximately the same volume dependence speaks against an increase in the density of states $N(E_f)$ as being responsible for the intrinsic T_c enhancement under pressure. In fact, early studies by us of the spin susceptibility of $\text{La}_{1.85}\text{Sr}_{0.15}\text{CuO}_4$ [64] and Y-123 [65] could detect no change whatsoever in $N(E_f)$ under pressure. From Eq. (1) it thus follows that the increase in T_c under pressure must come from an increase in the coupling strength \mathcal{V} itself.

To go further in the analysis it is necessary to ascertain what structural feature in the highly anisotropic oxides is responsible for the increase in \mathcal{V} under pressure.

What high pressure studies have taught us

Uniaxial pressure experiments provide information on the changes occurring in T_c for uniaxial stress applied both parallel and perpendicular to the CuO_2 planes. One of the most important experiments in this regard was carried out in 1997 by Meingast *et al.* [66] on Ca-substituted Y-123. These studies revealed that the compression along the c axis (perpendicular to the CuO_2 planes) has no effect on T_c , whereas compression in the CuO_2 planes caused the ubiquitous increase in T_c . A wealth of further uniaxial pressure experiments support this conclusion. This prompted Wijngaarden *et al.* [67] to state in 1999 that “Hence, there is quite some evidence that Δc mainly influences doping, while Δa mainly influences the intrinsic T_c ”. If we now use the available anisotropy compressibility data to convert the volume dependence $T_c \propto V^{-1.2}$ into a dependence on the in-plane lattice parameter a , we obtain

$$T_c \propto a^{-4.5}, \quad (5)$$

so that T_c is approximately proportional to the inverse square of the area of the CuO_2 planes. This is perhaps the single most important fundamental result of all high pressure experiments on the cuprate oxides. The message is: to further enhance T_c , try to find structures which are capable of *compressing* the CuO_2 planes without adding defects or increasing the buckling of these planes. Obviously this may not be easy to accomplish since these planes are quite stiff and it would be anticipated that structural attempts to compress the planes could easily cause them to buckle.

The next question is whether there is any correlation between the value of T_c at ambient pressure and the value of the in-plane lattice parameter a . In Fig. 4 it is seen that there is no such correlation. The single-plane material with the highest value of T_c (98 K) is Hg-1201 which has the largest value of a . In addition, the compound in Fig. 4 with the lowest value of T_c is $\text{La}_{1.85}\text{Sr}_{0.15}\text{CuO}_4$ which has the smallest value of a ! We note that applying 4 GPa pressure decreases a by about 1% which would generate an increase in T_c of about 6 K. Within that range of a in Fig. 4 the variation of T_c for the compounds listed is much greater. This simply means that the compounds presently available have not yet played the high-pressure card, i.e. they haven’t made use of the fact that T_c will increase appreciably if a is reduced. The system that managed to reduce a the most is $\text{La}_{1.85}\text{Sr}_{0.15}\text{CuO}_4$; perhaps the reason that its T_c value is the lowest of all systems in Fig. 4 arises from the strong structural distortions leading to considerable plane buckling.

To summarize, there are at least three principal ways that high external or lattice (chemical) pressures can be used to help maximize the value of T_c in the superconducting oxides:

1. Use pressure to reach optimal doping in system with few defects in or near the CuO_2 planes. This may be the best (only) way to reach optimal doping in systems with more than three CuO_2 planes.

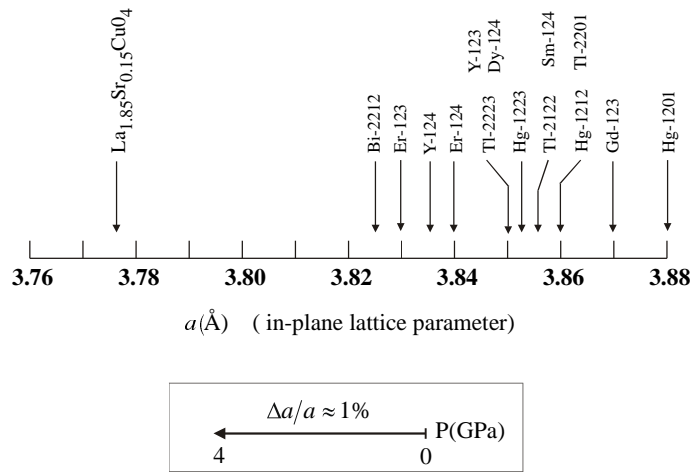


Fig. 4. Average lattice parameter in the CuO_2 plane for representative superconducting oxide systems at ambient pressure.

2. Use pressure to help flatten the CuO_2 planes.
3. Apply pressure to reduce the area of the CuO_2 planes, keeping them flat.
4. Use the result $T_c \propto a^{-4.5}$ to help identify the correct theoretical model and then apply this model to help further optimize the value of T_c . This will be the subject of a future paper.

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